

On Theories of the Grüneisen Parameter

by Steven B. Segletes and William P. Walters

ARL-TR-1303 March 1997

Cine) (Ambient Tiels Filled B

Approved for public release; distribution is unlimited.

19970224 127

The findings in this report are not to be construed as an official Department of the Army position unless so designated by other authorized documents.

Citation of manufacturer's or trade names does not constitute an official endorsement or approval of the use thereof.

Destroy this report when it is no longer need. Do not return it to the originator.

Army Research Laboratory

Aberdeen Proving Ground, MD 21005-5066

ARL-TR-1303 March 1997

On Theories of the Grüneisen Parameter

Steven B. Segletes, William P. Walters Weapons and Materials Research Directorate, ARL

Approved for public release; distribution is unlimited.

Abstract

A recently proposed equation of state is further examined for compatibility with prior theory. It is found, by adopting a more realistic functional relationship for one of the thermodynamic state variables and eliminating higher-order terms from the model, that three historical theories for determination of the Grüneisen function fall directly out of the recently proposed model. One is led to conclude that the newer model captures the essence of all of the older theories, each of which being applicable under an appropriate circumstance.

TABLE OF CONTENTS

	•	Page
	LIST OF FIGURES	. v
1.	BACKGROUND	. 1
2.	THE HIGH-COMPRESSION LIMIT FOR $\psi(V)$. 6
3.	A POWER LAW FOR $\psi(V)$. 8
4.	GRÜNEISEN MODELS	. 12
5.	NONIDEAL EFFECTS	. 16
6.	CONCLUSIONS	. 22
7.	REFERENCES	. 25
	DISTRIBUTION LIST	. 27
	REPORT DOCUMENTATION PAGE	. 43

INTENTIONALLY LEFT BLANK

LIST OF FIGURES

<u>Figure</u>		Page
1.	Limiting ψ values for several metals (solid lines), obtained by inserting TFD cold-	
	curve values into equation of state of Segletes. Superimposed (dashed lines) are	
	limiting ψ values associated with the linear ψ assumption	7
2.	Limiting ψ values for several metals (solid lines), obtained by inserting TFD cold-	
	curve values into equation of state of Segletes. Superimposed (dashed lines) are	
	limiting ψ values associated with the power-law $\psi(V)$ assumption	9
3.	Lattice binding energy as a function of nondimensionalized lattice spacing for a	
	material. Dark line is UCC model of Rose et al. [10]. Dashed lines are for model	
	of Segletes [15,16], with linear ψ assumption, for two particular values of $\Gamma_{\!_0}\psi'$	
	and an anharmonicity of 3	11
4.	Lattice binding energy as a function of nondimensionalized lattice spacing for a	
	material. Dark line is UCC model of Rose et al. [10]. Dashed lines (essentially	
	coincident) are for model of Segletes [15,16], with power-law $\psi(V)$ assumption,	
	for anharmonicity values of 2, 4, 6, 8, and 10	13
5.	Grüneisen function versus relative volume for hypothetical material with Γ_0 =2 and	
	K=1, using power-law $\psi(V)$, with x=4/9. Predictions using models of Slater [4],	
	Dugdale and MacDonald [5], and the free-volume theory [6] are shown for	
	comparison. Note that ambient value associated with power-law $\psi(V)$ matches	
	that of Slater	17

LIST OF FIGURES (CONTINUED)

Figure Property of the Propert	•	Page
6.	Grüneisen function versus relative volume for hypothetical material with Γ_0 =2 and K=1, using power-law $\psi(V)$, with x=2/3. Predictions using models of Slater [4],	
	Dugdale and MacDonald [5], and the free-volume theory [6] are shown for comparison. Note that ambient value associated with power-law $\psi(V)$ matches	
	that of Dugdale and MacDonald	17
7.	Grüneisen function versus relative volume for hypothetical material with Γ_0 =2 and	
	K=1, using power-law $\psi(V)$, with x=8/9. Predictions using models of Slater [4],	
	Dugdale and MacDonald [5], and the free-volume theory [6] are shown for	
	comparison. Note that ambient value associated with power-law $\psi(\boldsymbol{V})$ matches	
	that of the free-volume theory	18
8.	Pressure-volume curves for stainless steel. Cold data to 4 Mbar shown with filled	
	symbols. Hugoniot data to 2 Mbar shown with open symbols. Curves represent	
	cold-compression and shock-Hugoniot curves with $\Gamma\!\!=\!\!1.81$ and power-law $\psi(V)$	
	exponent x, equal to 0.72	20
9.	Grüneisen function versus relative volume for stainless steel with Γ_0 =1.81, using	
	power-law $\psi(V)$, with x=0.72. Predictions using models of Slater [4], Dugdale	
	and MacDonald [5], and the free-volume theory [6] are shown for comparison	20

1. BACKGROUND

The estimation and measurement of the Grüneisen function for metals is an important but difficult task. Simple formulas exist for its estimation, in terms of specific heat and compressibility, or alternately in terms of derivatives of the cold (*i.e.*, zero degree) compression curve, or still again in terms of derivatives of the lattice frequency. Depending on the validity of various assumptions about the material, some of these definitions may be redundant. However, when applied to actual substances, these methods can produce estimates that vary by significant amounts (100%, in some cases [1]), even at ambient conditions.

Some variation is due to the fact that the Grüneisen assumption, which has the Grüneisen function independent of temperature, is only an approximation. It is well known [2,3], for example, that at low temperatures (at fractions of the Debye temperature) the Grüneisen parameter is indeed temperature dependent. Grodzka [1] presents data showing a significant thermal variation in the Grüneisen parameter at higher temperatures as well, for some materials. Other variations in the estimate arise out of the differences in modeling of the propagation velocities associated with lattice vibration. Three widely referenced models of this variety, which provide not only an ambient value but also a volume-dependent estimate of the Grüneisen function, are those of Slater [4], Dugdale and MacDonald [5], and the free-volume theory [6]. Pastine [7] and Vashchenko and Zubarev [6] have attempted to generalize results by showing that the variation of Poisson's ratio with volume can account for some of the differences in the various models of the Grüneisen parameter. Unfortunately, experimental knowledge of how Poisson's ratio varies with compression and/or temperature is as elusive as knowledge of the Grüneisen function itself.

Historically, the estimation of the Grüneisen function was further complicated by the paucity of cold-compression data. Available high-pressure data were almost exclusively that of shock transition, and so the cold-compression curves, needed by the various models in order to estimate the Grüneisen function, were necessarily back-extrapolated from shock data using a Grüneisen function that was a priori unknown (e.g., see Walsh et al. [8]).

To some extent, this uncertainty in the functional behavior of the Grüneisen function has persisted to the present. For example, Segletes [9] examined the Grüneisen formulations present in five of the widely used hydrocodes, which were developed by the U.S. Department of Energy (DOE) national labs and the U.S. Department of Defense (DOD), as well as industry. In examining these codes, which are utilized to model the high-compression deformations associated with high-velocity impact and/or explosive loading, it was noted that a common Grüneisen formulation was not employed. Not only were the Grüneisen formulations different from code to code, but Segletes was able to show, in some cases, that thermodynamic instabilities would result because of an incompatible selection of Grüneisen and Hugoniot functions.

In a significant work, which, on the surface, might not appear to shed light on the problem of estimating the Grüneisen function, Rose *et al.* [10] proposed (in 1984) a universal cold curve, which was able to provide a generic function for the lattice binding energy, which could be appropriately scaled to match theoretical and experimental data for many metals. Since the work of Rose and his colleagues, others have continued in the area as well, notably the same team led by Vinet [11,12] and, more recently, a group led by Baonza [13,14]. These later efforts succeeded in introducing temperature effects and thus produced a Grüneisen-style equation of state. However, in all of these more recent efforts cited, the function governing the reference-compression curve was mutually exclusive of the function used to model the thermal effects. In the works of Vinet *et al.*, the variation of the bulk modulus was fitted for a number of materials, which effectively parlayed into a Grüneisen function. In the case of Baonza *et al.*, a particular, commonly employed, functional form of the Grüneisen function was used—namely, Γ/V equals a fitted constant. While all of these works are worthy efforts in their own right, their primary contribution lies in an area other than the theory of the Grüneisen function.

Segletes [15,16] recently proposed an equation of state for metals, which captures, in the absence of phase change, the behavior of metals into the megabar range. The model was likewise inspired by the universal cold-curve work of Rose et al. [10]. However, unlike the efforts of Vinet et al. [11,12] and Baonza et al. [13,14], in which the function governing the cold compression was mutually exclusive of the function used to model the thermal effects, the recent

work of Segletes, by expressing the binding-energy function in terms of lattice frequency, as opposed to lattice compression, unified the functions governing cold-compression and thermal effects. In essence, a specification of the cold-compression behavior in Segletes' model automatically yields the Grüneisen function, and conversely, a specification of the thermal effects, by way of the Grüneisen function, automatically yields a cold-compression curve.

Segletes' equation is of the Grüneisen variety and is given by

$$p \psi - E = E_b \left\{ \left[(\Theta/\Theta_0)^K - 1 \right] + K(K - 1) (\Theta/\Theta_0)^K \ln(\Theta/\Theta_0) \right\} , \qquad (1)$$

where p and E are the pressure and specific internal energy, respectively, and the constant E_b is the specific binding energy of the lattice. The terms Θ_0 and Θ are the ambient and current values of the characteristic temperature of the lattice, which, according to the tenets of Grüneisen theory, is assumed to be a function of volume only and is proportional to the maximum lattice vibrational frequency, v. The concepts of characteristic temperature and lattice frequency derive from the early work on specific heats at low temperatures, by Einstein, Debye, Born and von Karman, and others. In the current context, the characteristic temperature serves as a macroscopic parameter that permits the vibrational frequency of the lattice (a microscopic quantity) to be inferred—namely, the theory of specific heats has these two parameters directly proportional. Returning to eqn. (1), the parameter ψ is the thermodynamic state variable equal to the ratio of specific volume to the Grüneisen parameter, V/Γ , introduced by Segletes [17,18] for convenience in manipulating the thermodynamic equations. The constant K is given by

$$K = \frac{C_0}{\Gamma_0 \sqrt{E_h}} \quad , \tag{2}$$

which incidentally is $1/(3\Gamma_0)$ times the anharmonicity of the lattice. Values for K have been observed to cluster in the 2/3 to 4/3 range for metals, which is compatible with the model's

behavior becoming greatly idealized (i.e., linear in Θ/Θ_0) when the constant K takes on a value of unity. C_0 is the bulk sound speed at absolute zero temperature and pressure, and Γ_0 is the ambient value of the Grüneisen parameter.

In this model, the cold curve (i.e., zero-degree isotherm) is given by

$$E_c = E_b \left\{ 1 - \left[1 - K \ln(\Theta/\Theta_0) \right] (\Theta/\Theta_0)^K \right\}$$
 (3)

and

$$p_c = \frac{E_b K^2}{\Psi} (\Theta/\Theta_0)^K \ln(\Theta/\Theta_0) , \qquad (4)$$

where p_c and E_c are the cold pressure and specific cold energy, respectively.

In macroscopic terms, the Grüneisen function, Γ , is defined by $\Gamma = V/\psi = V(\partial p/\partial E)_V$. From microscopic arguments, however, the characteristic temperature, Θ (alternately, the maximum lattice vibrational frequency, ν), is related to the ψ variable through the relationship that defines the Grüneisen parameter:

$$\frac{\mathbf{v}'}{\mathbf{v}} = \frac{\Theta'}{\Theta} = -\frac{1}{\mathbf{w}} , \qquad (5)$$

where primes (here and throughout the paper) denote ordinary differentiation with respect to specific volume. In the general case, the differentiation in eqn. (5) would be partial at constant temperature. However, for Grüneisen materials, in which $\Gamma = \Gamma(V)$ only, the derivative in eqn. (5) becomes ordinary.

It is interesting to note the distinction between Segletes' equations and those of an ideal, harmonic oscillator, which is often used to approximate a crystal lattice for purposes of theoretical derivation. In the case of Planck's harmonic oscillator at zero temperature [19], dE_c/dv is constant. No lattice is truly harmonic, of course, or else its thermal expansion coefficient and Grüneisen function would both be identically zero. From eqns. (3) and (5), it can be seen, for the idealized case where K is unity, that the current model would have $dE_c/dv \approx \ln(v/v_0)$. A theoretical explanation for this distinction is not yet postulated.

Once the ψ function is defined, integration of eqn. (5) gives the characteristic temperature, which is the primary equation-of-state variable in eqn. (1). In the original papers, Segletes [15,16] incorporated a linear relationship for $\psi(V)$, based on results of his prior work on thermodynamic stability [9,17,18], given in general form by

$$\psi/\psi_0 = 1 - (\Gamma_0 \psi') (1 - V/V_0) . \tag{6}$$

The ambient values of specific volume and Grüneisen parameter determine ψ_0 , while the constant $(\Gamma_0 \psi')$ was the sole material parameter that was actually fitted to the cold curve of each metal. The fitted values of $(\Gamma_0 \psi')$, for nine metals, ranged from 0.43 to 0.88, with a vast majority clustered in the vicinity of 0.8. In terms of compression, μ , equal to $(V_0 - V)/V$, the linear ψ translates to a Grüneisen function of the form $\Gamma = \Gamma_0/(1+\beta\mu)$, with β as the fitting parameter.

This assumed linear relationship produced an excellent match to empirical data and first-principles computations over a very wide range of specific volumes. However, for extremely large compressions, this linear relationship can no longer be expected to hold, as it is later shown that the ψ variable must vanish in the high-compression limit $(V \to 0)$, whereas the linear form of eqn. (6) will not do so for values of $(\Gamma_0 \psi')$ other than unity.

In this report, a power law for the $\psi(V)$ parameter, to replace the linear relationship originally employed, is adopted. Though this choice of function still suffers deficiencies at very large compressions, a number of positive outcomes result in the context of Segletes' model [15,16] with this change: 1) the $\psi(V)$ parameter vanishes with diminishing specific volume, as is required; 2) binding-energy expansion data for actual metals are better fit with the power law, as compared to the linear assumption; and most importantly, 3) Segletes' equation of state is shown, by neglecting higher-order terms, to reduce to the model of Slater [4], or Dugdale and MacDonald [5], or to the free-volume theory [6], depending on the value of the exponent used in the power law for $\psi(V)$.

2. THE HIGH-COMPRESSION LIMIT FOR $\psi(V)$

With the linear relation for $\psi(V)$, given by eqn. (6), the high-compression limit for ψ approaches a value of ψ_0 (1 – $\Gamma_0\psi'$), which would typically lie in the $0.2 \cdot \psi_0$ range for metals. To show what high-pressure limit is appropriate for ψ , in the context of the new model, theoretical values for cold pressure and energy using Thomas-Fermi-Dirac (TFD) theory for several metals were inserted into the new model in order to back out values for ψ . In this manner, the high-pressure limiting functional behavior for ψ , which would be required to match TFD theory, was obtained. Values for the TFD data were obtained using the analytical fitting equations described by Kerley [20].

It is not asserted here that the new equation of state can or should capture the high-compression atomic interactions modeled with TFD zero-temperature theory. Rather, it merely makes sense, from an application point of view, that the current model should blend smoothly into TFD theory, so that a computational implementation can smoothly transition between applicable theories.

Results of the TFD fits for ψ are shown in Figure 1, for several metals. Superimposed on the graph are lines of constant $\Gamma_0 \psi'$, which characterized the original linear form employed by Segletes [15,16]. A finite limiting value of ψ implies that Γ approaches zero in the high-

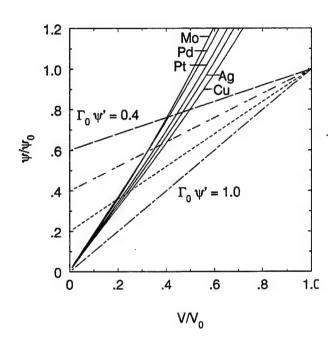


Figure 1. Limiting ψ values for several metals (solid lines), obtained by inserting TFD cold-curve values into equation of state of Segletes. Superimposed (dashed lines) are limiting ψ values associated with the linear ψ assumption.

compression limit, which is known not to be the case. The TFD results indicate that ψ must drop to zero with decreasing specific volume, which indicates that the limiting value of Γ is nonzero, but instead related to the limiting slope. Regardless of material, the TFD results are roughly linear, having a slope of approximately 2 in the very-low-volume limit, with the slope diminishing slightly with increasing volume. A finite limiting slope of value m in Figure 1 implies a limiting value of the Grüneisen parameter of Γ_0/m . The fact that (in the case of $m\approx 2$) $\Gamma_0/2$ doesn't equal 1/2 or 2/3, as might be expected, merely indicates that the current model fails to capture the essence of TFD theory, which is no great surprise since the model's equations are not those of TFD theory. In any event, the linear treatment of ψ , originally employed, does not blend smoothly into the realm where TFD theory governs. Another functional choice for ψ might be more appropriate.

3. A POWER LAW FOR $\psi(V)$

As an alternative to the linear relationship for $\psi(V)$, given by eqn. (6), a power law of the form

$$\psi/\psi_0 = (V/V_0)^x \tag{7}$$

is investigated. The power-law form for $\psi(V)$ retains the mathematical simplicity necessary to manipulate the governing equations conveniently, and despite a few deficiencies that will be pointed out, it seems to be a step in the right direction. In terms of compression, μ , the power-law ψ translates to a Grüneisen function of the form $\Gamma = \Gamma_0/(1+\mu)^{1-x}$.

Figure 2 shows the same TFD results as presented previously, except that several contours of a power-law $\psi(V)$ are superimposed on the graph, rather than the linear contours of Figure 1. At first glance, the power-law $\psi(V)$ might seem to offer only minimal benefit over the linearized form. Both forms cross over, rather than asymptote, to the TFD curves. Also, the power-law curves, though reducing to a value of zero, as is required, do so with an infinite limiting slope, so that the limiting value of Γ is not positive, but remains zero. On the positive side, however,

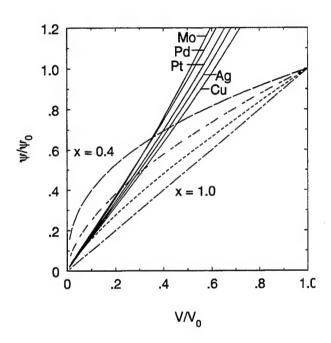


Figure 2. Limiting ψ values for several metals (solid lines), obtained by inserting TFD cold-curve values into equation of state of Segletes. Superimposed (dashed lines) are limiting ψ values associated with the power-law $\psi(V)$ assumption.

the curve corresponding to an exponent of 0.8 does blend nicely into the TFD curves. And of the nine metals examined by Segletes [15,16], seven had values for $\Gamma_0 \psi'$ (corresponding roughly to exponent x) in the 0.75 to 0.88 range, where the power law fit approximates the TFD asymptote very well.

As far as actual cold-compression data, the power law and the linear assumption for $\psi(V)$ are nearly identical for $\Gamma_0 \psi'$ values of 0.8 over relative volumes from unity down to 0.5. In this region, the power law provides a slightly more concave-upward curvature to the cold curve, which can be mitigated by choosing a value of exponent x, slightly below the fitted value of $\Gamma_0 \psi'$ (e.g., whereas $\Gamma_0 \psi'$ =0.76 fits stainless steel, a value of x=0.72 fits comparably well). Over the range of available cold-compression data [10,21,22], the power-law and linear $\psi(V)$ formulations can be made virtually indistinguishable with appropriate selection of x, both resulting in excellent fits to the data.

On the expansion end of the curve, the power-law $\psi(V)$ is seen to provide an improved correlation to data. At high values of anharmonicity (~9), the linear model was able to match the universal cold-curve (UCC) fit of Rose *et al.* [10] with a value of $\Gamma_0 \psi'$ approaching unity. The power law does equally well here, since for an exponent approaching unity, the linear and power forms are identical. However, for low anharmonicities (~3), no single value of $\Gamma_0 \psi'$ could produce a match to the UCC model [15,16], but instead required a slope that was initially around 0.8, reducing to 0.55 at the larger expansions. Figure 3 compares the lattice potential energy predicted by the UCC model of Rose to that using the linear- ψ approximation. In this graph, the abscissa, a, represents the nondimensional, relative lattice spacing employed by Rose *et al.* [10].

By contrast, the power law $\psi(V)$ is able to match the UCC model closely with a single value of exponent x for a given anharmonicity. The value of the exponent x, which will provide a match to the UCC model, is given approximately by

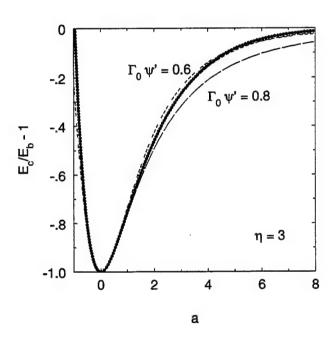


Figure 3. Lattice binding energy as a function of nondimensionalized lattice spacing for a material. Dark line is UCC model of Rose et al. [10] Dashed lines are for model of Segletes [15,16], with linear ψ assumption, for two particular values of $\Gamma_0 \psi'$ and an anharmonicity of 3.

$$x = \frac{2}{3} + \frac{1}{3} \cdot (\eta/10) , \qquad (8)$$

where η is the value of anharmonicity, as defined by Rose *et al.* [10]. The magnitude of anharmonicity provides a metric on the deviation of an actual crystal's behavior from the harmonic idealization, in which the binding force of the atomic lattice would be proportional to the atom's displacement from the rest position. It is later shown that Segletes' model [15,16] approximates the harmonic solid described by Dugdale and MacDonald [5] when the value of exponent x is selected as 2/3, in accordance with the lower limit of eqn. (8), as $\eta \rightarrow 0$. Figure 4 compares the lattice binding energy predicted by the UCC model to that of the current model, when a power-law $\psi(V)$ is employed, the exponent being selected in accordance with eqn. (8). Though not labeled, the dark curve represents the UCC model, while the other five, essentially overlapping, curves are for the current model with anharmonicities of 2, 4, 6, 8, and 10.

4. GRÜNEISEN MODELS

Certain mathematical consequences of a power-law $\psi(V)$, eqn. (7), which are subsequently used, are easily derived and are simply stated as follows:

$$(\Gamma_0 \psi')_{V_0} = x \quad ; \tag{9}$$

$$\psi' = x \cdot (\psi/V) \quad ; \tag{10}$$

$$\psi' - n \cdot (\psi/V) = (x - n) \cdot (\psi/V) \quad ; \quad \text{and}$$
 (11)

$$\psi \left[\psi'' - n \cdot (\psi/V)' \right] = (x - n) (x - 1) \cdot (\psi/V)^2 . \tag{12}$$

We now intend to show the primary result of this paper—namely, that three historical Grüneisen formulations (those of Slater [4], Dugdale and MacDonald [5], and the free-volume

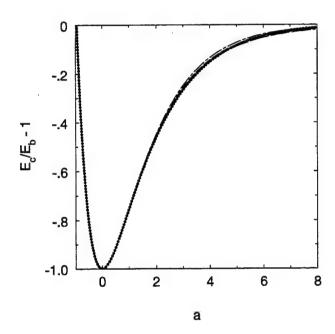


Figure 4. Lattice binding energy as a function of nondimensionalized lattice spacing for a material. Dark line is UCC model of Rose et al. [10]. Dashed lines (essentially coincident) are for model of Segletes [15,16], with power-law $\psi(V)$ assumption, for anharmonicity values of 2, 4, 6, 8, and 10.

theory [6]) may be recovered through manipulation of the current model, if the material is idealized and higher-order terms are neglected. Start with eqn. (4) and multiply by a power of specific volume, V^n , where n is, at this point, still an arbitrary constant:

$$(p_c V^n) = \frac{E_b K^2 V^n}{\Psi} (\Theta/\Theta_0)^K \ln(\Theta/\Theta_0) . \qquad (13)$$

The first and second derivatives of this equation may be obtained in a tedious but straightforward manner as:

$$(p_c V^n)' = -\frac{E_b K^2 V^n}{\psi^2} (\Theta/\Theta_0)^K \left\{ [K + \psi' - n(\psi/V)] \ln(\Theta/\Theta_0) + 1 \right\}$$
 (14)

and

$$(p_c V^n)'' = \frac{E_b K^2 V^n}{\Psi^3} (\Theta/\Theta_0)^K \left([K + 2\psi' - n(\psi/V)] \left\{ [K + \psi' - n(\psi/V)] \ln(\Theta/\Theta_0) + 1 \right\} + \dots$$

$$[K + \psi' - n(\psi/V)] - \psi[\psi'' - n(\psi/V)'] \ln(\Theta/\Theta_0) \right)$$
(15)

Take eqn. (15) multiplied by $(-\psi)$ and divide by eqn. (14) to obtain:

$$-\psi \frac{(p_c V^n)''}{(p_c V^n)'} = 2K + 3\psi' - 2n(\psi/V) - \frac{\left\{ [K + \psi' - n(\psi/V)]^2 + \psi [\psi'' - n(\psi/V)'] \right\} \ln(\Theta/\Theta_0)}{1 + [K + \psi' - n(\psi/V)] \ln(\Theta/\Theta_0)} . (16)$$

Eqns. (10–12) may be substituted into this expression to eliminate ψ' , the left hand ψ may be broken out as V/Γ , and the equation may be algebraically manipulated to give

$$-\frac{3x-2n}{2} - \frac{V}{2} \frac{(p_c V^n)''}{(p_c V^n)'} = \Gamma \times \left(K - \frac{1}{2} \frac{\left\{ [K + (x-n)(\psi/V)]^2 + (x-n)(x-1)(\psi/V)^2 \right\} \ln(\Theta/\Theta_0)}{1 + [K + (x-n)(\psi/V)] \ln(\Theta/\Theta_0)}\right). \tag{17}$$

The large term in parentheses at the end of this equation takes on the value K at ambient conditions and diminishes with compression as the logarithm terms exert their influence. The material parameter K, for materials studied by Segletes [15,16], has generally been close to the idealized value of unity [whereupon $(C_0/\Gamma_0)^2=E_b$]. For the sake of discussion, therefore, consider the large term in parentheses as unity, as an approximation to the situation near ambient volume $[\ln(\Theta/\Theta_0)\approx 0]$ for an idealized material $(K\approx 1)$. Eqn. (17) becomes

$$\Gamma = -\frac{3x - 2n}{2} - \frac{V}{2} \frac{(p_c V^n)''}{(p_c V^n)'} . \tag{18}$$

In this equation, x, which denotes the power of $\psi(V)$, is a material parameter. If we select the value of parameter n, which has been arbitrary to this point, to be related to x in the following way:

$$n = 3x - 4/3 (19)$$

then x may be eliminated from eqn. (18) to yield

$$\Gamma = -\frac{4 - 3n}{6} - \frac{V}{2} \frac{(p_c V^n)^{n/2}}{(p_c V^n)^{n/2}}.$$
 (20)

Eqn. (20) is precisely the generalized result presented by Vaschenko and Zubarev [6] for the

Grüneisen function. In this equation, if n=0 (i.e., when x=4/9), Slater's [4] result is obtained. If n=2/3 (i.e., when x=2/3), then the result of Dugdale and MacDonald [5] is recovered. Finally, for the case where n=4/3 (i.e., when x=8/9), the free-volume theory [6] expression follows. What this result tells us is that any one of the three historical Grüneisen models may be nominally applicable, depending on the properties of the material of interest. In the case of those metals already studied by Segletes [15,16], values for $\Gamma_0 \psi'$, which correspond to exponent x at ambient conditions [via eqn. (9)], fell as low as 0.43 ($\approx 4/9$) for molybdenum and as high as 0.88 ($\approx 8/9$) for rubidium, with the vast majority of materials in the 0.75-0.85 range.

5. NONIDEAL EFFECTS

Let us now consider the practical influence of the large term in parentheses in eqn. (17), idealized as unity in the previous section. The leading K term in the braces affects the ambient value of the Grüneisen parameter, while the remaining term involving logarithms governs the rate at which the Grüneisen function changes with volume. If Segletes' equation of state is accepted, then the fact that the large term in parentheses diminishes with volume indicates that all of the historical Grüneisen formulations (Slater, Dugdale-MacDonald, and free-volume theory) will tend to overestimate the rate of change of the Grüneisen function with compression when using a cold-compression curve as the baseline.

Whereas the historical Grüneisen models are based on various theories of lattice vibration, the model of Segletes [15,16], in contrast, is shown to fit actual cold and Hugoniot data well to several megabars of pressure. Thus, if one accepts the validity of the Grüneisen assumption $[\Gamma = f(V)]$, then, over the compression range of actual thermodynamic data to which Segletes' model fits well, one must accept as more accurate the mildly varying Γ of that model to the more widely varying estimates produced by the theories of Slater, Dugdale-MacDonald and/or the free-volume theory.

To see how the models compare, Figures 5–7 depict the Grüneisen functional behavior for a hypothetical material with parameter K=1 and $\Gamma_0=2$. These three figures depict four curves

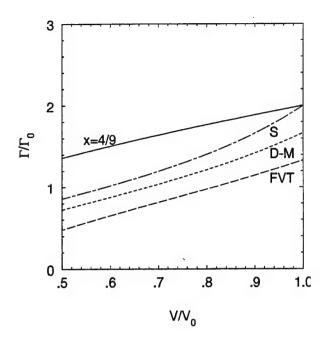


Figure 5. Grüneisen function versus relative volume for hypothetical material with Γ_0 =2 and K=1, using power-law $\psi(V)$, with x=4/9. Predictions using models of Slater [4], Dugdale and MacDonald [5], and the free-volume theory [6] are shown for comparison. Note that ambient value associated with power-law $\psi(V)$ matches that of Slater.

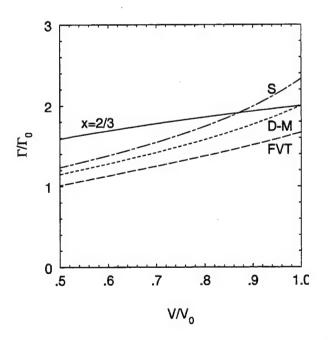


Figure 6. Grüneisen function versus relative volume for hypothetical material with $\Gamma_0=2$ and K=1, using power-law $\psi(V)$, with x=2/3. Predictions using models of Slater [4], Dugdale and MacDonald [5], and the free-volume theory [6] are shown for comparison. Note that ambient value associated with power-law $\psi(V)$ matches that of Dugdale and MacDonald.

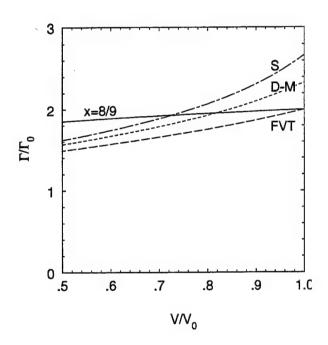


Figure 7. Grüneisen function versus relative volume for hypothetical material with $\Gamma_0=2$ and K=1, using power-law $\psi(V)$, with x=8/9. Predictions using models of Slater [4], Dugdale and MacDonald [5], and the free-volume theory [6] are shown for comparison. Note that ambient value associated with power-law $\psi(V)$ matches that of the free-volume theory.

each. In each graph, there is a Grüneisen curve corresponding to the power-law $\psi(V)$, with exponent x equal to 4/9, 2/3, and 8/9 respectively. The resulting cold curve from the current model is used to show what the models of Slater, Dugdale-MacDonald, and the free-volume theory would predict for the behavior of the Grüneisen function, depicted in each of the figures by curves labeled S, D-M, and FVT, respectively. Per eqn. (17), with K equal to unity, the ambient values for the current thory match the value of Slater when x=4/9, that of Dugdale and MacDonald when x=2/3, and that of the free-volume theory when x=8/9.

To see a comparison of the other models to that of Segletes, for an actual material, we consider the case of stainless steel, because of the availability of good cold-curve data out to 4 Mbar. Figure 8 shows the cold curve and Hugoniot for the steel. The data points are the same as those originally analyzed by Segletes [15,16]. However, unlike the earlier work, in which a linear $\psi(V)$ function was fit to the data with a slope of 0.76, the cold curve is fit on this occasion with a power law, with an exponent, x, of 0.72. The Hugoniot fit is excellent as far as data are available (out to 2 Mbar). If the Grüneisen assumption is accepted, then a power law exponent of 0.72 must provide an accurate reflection of the Grüneisen function down to a relative volume of 0.68 (where Hugoniot data stop), and most likely provides an accurate reflection of the function down to a relative volume of 0.55 (where cold data stop). Figure 9 shows the Grüneisen function corresponding to this fit with the solid curve. Overlaid on the figure are the results of the models of Slater, Dugdale and MacDonald, and the free-volume theory, which are derived from the curvature of the stainless steel cold-curve fit. As pointed out previously, the other theories all show a Grüneisen function that decreases more rapidly in compression than the current model and data suggest.

Of course, at very high compressions and temperatures, electronic and non-Grüneisen thermal effects will come into play, and so even if a simple power law for $\psi(V)$ might blend into the TFD cold curve, it will likely fail to capture the high-temperature thermal effects properly. For example, the shock Hugoniot of most materials asymptotes at a V/V_0 value of around 0.2. Such an asymptote corresponds to a relative compression, μ , of 4. Segletes [9] proved that the asymptotic value of the Grüneisen parameter on the Hugoniot cannot exceed (and, in fact, must

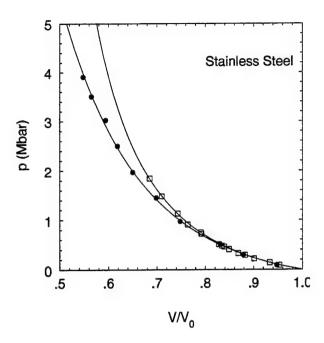


Figure 8. Pressure-volume curves for stainless steel. Cold data to 4 Mbar shown with filled symbols. Hugoniot data to 2 Mbar shown with open symbols. Curves represent cold-compression and shock-Hugoniot curves with Γ =1.81 and power-law Ψ (V) exponent x, equal to 0.72.

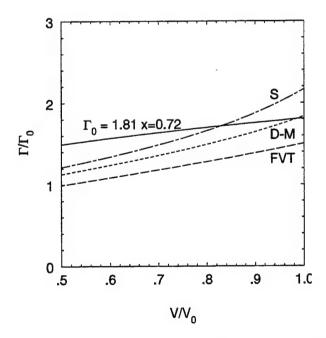


Figure 9. Grüneisen function versus relative volume for stainless steel with $\Gamma_0 = 1.81$, using power-law $\psi(V)$, with x=0.72. Predictions using models of Slater [4], Dugdale and MacDonald [5], and the free-volume theory [6] are shown for comparison.

take on) a value of $2/\mu_{asym}$, which in this case, gives a value of 0.5. If the ambient value of the Grüneisen parameter were a typical value of Γ_0 =2, then this asymptotic shock limit would possess a ψ/ψ_0 value of 0.8 at a V/V_0 value of 0.2. A quick look at the power-law curves of Figure 2 reveals that power-law $\psi(V)$ functions with exponents between 0.4 and 1.0 will not reach the point (0.2, 0.8). Thus, it would seem that the only way the Hugoniot can get to the point (0.2, 0.8) while the cold curve asymptotes to the TFD curves is if ψ is temperature dependent.

There is some reason to suspect such temperature-dependent behavior, by looking, for example, at the aluminum Hugoniot fit of Segletes [15,16]. When using the known ambient Grüneisen value (Segletes used 2.39) with the current model, the Hugoniot prediction was quite poor. But by using an ambient value of half that amount, which doubles the value of ψ, the aluminum Hugoniot and cold curve were fit with precision to 9 Mbar ($V/V_0 = 0.4$). One might be led to conclude that w is, indeed, temperature dependent and that, in the case of aluminum, a dependence occurs at relatively low temperatures. Such a thermal dependency would be able to explain how the Hugoniot can achieve relatively large asymptotic values of ψ/ψ_0 , while, at the same time, the cold-curve value diminishes below this level. A significant thermal dependency of the Grüneisen function for aluminum has been reported by Grodzka [1]. In particular, he cites experimental work with shocked aluminum foam that shows the ambient-volume aluminum Grüneisen parameter to decrease from a cold value of 2.5 to a value below 1.4 at 6500 J/g of specific internal energy. Using a value of 3R/W to approximate the specific heat over this range (where R is the universal constant, and W is the atomic weight of aluminum), 6500 J/gcorresponds to a temperature of approximately 7000 K, and would result from a moderatestrength shock in the 1.5 Mbar range.

As a final note, another way in which Segletes' model and those of Slater, Dugdale and MacDonald, and the free-volume theory may be contrasted is accomplished by rearranging eqn. (20) as

$$\left[2 + \frac{4 - 3n}{3} (\psi/V)\right] \psi(p_c V^n)' + \psi^2(p_c V^n)'' = 0 .$$
 (21)

As before, the value of n determines which of the three older models is being discussed. This equation is first-order in $(p_c V^n)'$. By comparison, the current model, with a power-law $\psi(V)$ function, satisfies the following identity:

$$K^{2}(p_{c}V^{x}) + \left[2K + x(\psi/V)\right]\psi(p_{c}V^{x})' + \psi^{2}(p_{c}V^{x})'' = 0 , \qquad (22)$$

which is second-order in p_cV^x . These two equations are obviously different, in that eqn. (21) contains only derivatives of p_cV^n , while eqn. (22) involves both p_cV^x and its derivatives. However, there is also a great deal of similarity in that not only are both equations hyperbolic, but for the idealized, harmonic case of K = 1 and x = n = 2/3, eqns. (21) and (22) differ only by the first term of eqn. (22).

If eqn. (19) is used to convert eqn. (21) from n-based to x-based (to consider cases when $n \neq x$), it can be shown (for K=1) that the transformed eqn. (21) also differs from eqn. (22) only in the first term of eqn. (22), regardless of the value of x. It is for for this reason that at ambient conditions, when p_c [and thus the first term of eqn. (22)] is zero, the value of Γ_0 from Segletes' model matches the older theories as a function of x, respectively. We can thus think of the older theories as each being applicable for a particular class of material (defined by material parameter x), but at the same time neglecting the p_cV^x term necessary to capture the empirical data correctly.

6. CONCLUSIONS

In this report, the recently proposed equation of state of Segletes [15,16] has been examined for compatibility with prior theory. In the original work, an excellent match to available data and first-principles computations was achieved with the use of an assumed-linear expression for the $\psi(V)$ function, defined as V/Γ . In the current work, the use of a power law for $\psi(V)$ was examined in lieu of the linear relationship.

A number of positive outcomes resulted with the use of this power law. First, the low-volume limit for $\psi(V)$ approaches zero, which was shown to be desirable in order to have the current theory blend smoothly into Thomas-Fermi-Dirac cold-compression theory. Secondly, over the range of available cold-compression data, the linear and power-law fits can be made virtually indistinguishable, both providing excellent fits to the data. Finally, the binding energy function using a power-law $\psi(V)$ assumption was able to reproduce the fit of Rose *et al.* [10] better, especially in the area of lattice expansion. In this regard, an expression given by eqn. (8) shows a simple relation between material anharmonicity and the fitted parameter of the current model, in order to match the results of Rose *et al.* closely.

The primary result of the paper has been in showing how Segletes' model with a power-law $\psi(V)$, for the case of an idealized material, ignoring higher-order terms, is able to reproduce three historical Grüneisen function models, depending on the value of the fitted material parameter in Segletes' model. In particular, when the model parameter, which relates to the decrease of the Grüneisen function with compression, takes on a value of 4/9, the result of Slater [4] is obtained. When the parameter is taken as 2/3, the model of Dugdale and MacDonald [5] is recovered. Finally, if the model parameter takes the value of 8/9, the free-volume theory [6] expression follows.

The paper then has shown how Segletes' model differs from the historical Grüneisen models, if the higher-order terms are retained. It becomes apparent that the historical models of the Grüneisen-function predict a more rapid decrease of the Grüneisen function with compression than does the current model. Some speculations have been put forward here on the possibility of a temperature-dependent ψ function, which would, by definition, violate the Grüneisen hypothesis. Not only would this possibility help to explain some seeming-incompatibilities between the behaviors of the high-compression cold curve and the asymptotic Hugoniot, but a temperature-dependent ψ could also explain the otherwise anomalous behavior for aluminum cited by Segletes [15,16] in the relative volume range of 0.85 to unity. Finally, the older Grüneisen models and Segletes' model have been mathematically expressed in a different form, which clearly shows the latter's distinction from the others. In the case of the comparison

between an idealized, harmonic material in the current model to the model of Dugdale and MacDonald, the distinction originates from a single $pV^{2/3}$ term, which represents a higher-order departure from the earlier theories.

There are several future efforts in this area that would be of utility. From a position of theoretical interest, it would be useful to reconcile the current theory with the very-high-pressure, equation-of-state theories, in something more than an empirical blend. Barring this reconciliation, however, the current theory has still shown its applicability into the megabar regime with excellent results, and should be of utility to those modeling material deformation resulting from impact and conventional explosive events in the ordnance regime. Also, the area of non-Grüneisen (and/or phase change) effects seems an interesting and important area for further work. For many materials, the Grüneisen approach seems perfectly adequate over a large range of compressions and pressures. However, there do seem to be materials (aluminum has been mentioned in this paper) for which non-Grüneisen effects at moderate pressures might provide an adequate explanation.

7. REFERENCES

- Grodzka, P.G. "Grüneisen Parameter Study." LMSC/HREC A784868, Lockheed Missile Space Company, Huntsville, AL, October, 1967.
- 2. Blackman, M. "The Specific Heat of Solids." In *Encyclopedia of Physics; Volume VII* Part 1: Crystal Physics I. S. Flügge (ed.), Berlin: Springer-Verlag, pp. 325–382, 1955.
- 3. Barron, H.T.K. "On the Thermal Expansion of Solids at Low Temperatures." *Philosophical Magazine*, **40**, pp. 720–734, 1955.
- 4. Slater, J.C. Introduction to Chemical Physics. New York: McGraw Hill, 1939.
- 5. Dugdale, J.S., and D.K.C. McDonald. "The Thermal Expansion of Solids." *Physical Review*, **89** (4), pp. 832–834, 1953.
- 6. Vashchenko, V.Y., and V.N. Zubarev. "Concerning the Grüneisen Constant." *Soviet Physics—Solid State*, **5** (3), pp. 653–655, 1963.
- 7. Pastine, D.J. "Formulation of the Grüneisen Parameter for Monatomic Cubic Crystals." *Physical Review*, **138** (3A), pp. A767–A770, 1965.
- 8. Walsh, J.M., M.H. Rice, R.G. McQueen, and F.L. Yarger. "Shock-Wave Compressions of Twenty-Seven Metals. Equations of State of Metals." *Physical Review*, **108** (2), pp. 196–216, 1957.
- 9. Segletes, S.B. "Thermodynamic Stability of the Mie-Grüneisen Equation of State, and Its Relevance to Hydrocode Computations." *Journal of Applied Physics*, **70** (5), pp. 2489–2499, 1991. Erratum: **71** (2), p. 1074.
- 10. Rose, J.H., J.R. Smith, F. Guinea, and J. Ferrante. "Universal Features of the Equation of State of Metals." *Physical Review B*, **29** (6), pp. 2963–2969, 1984.
- 11. Vinet, P., J.R. Smith, J. Ferrante, and J.H. Rose. "Temperature Effects on the Universal Equation of State for Solids." *Physical Review B*, **35** (4), pp. 1945–1953, 1987.

- 12. Vinet, P., J.H. Rose, J. Ferrante, J.R. Smith. "Universal Features of the Equation of State of Solids." *Journal of Physics: Condensed Matter*, 1, pp. 1941–1963, 1989.
- 13. Baonza, V.G., M. Cáceres, and J. Núnez. "Universal Compressibility Behavior of Dense Phases." *Physical Review B*, **51** (1), pp. 28–37, 1995.
- 14. Baonza, V.G., M. Taravillo, M. Cáceres, J. Núnez. "Universal Features of the Equation of State of Solids from a Pseudospinodal Hypothesis." *Physical Review B*, **53** (9), pp. 5252–5258, 1996.
- 15. Segletes, S.B. "An Equation of State for Metals." ARL-TR-1270, U.S. Army Research Laboratory, Aberdeen Proving Ground, MD, November, 1996.
- 16. Segletes, S.B. "An Equation of State for Metals." *Journal of Applied Physics*, submitted for review, 1996.
- 17. Segletes, S.B. "Further Examinations on the Thermodynamic Stability of the Mie-Grüneisen Equation of State." *Journal of Applied Physics*, **76** (8), pp. 4560–4566, 1994.
- 18. Segletes, S.B. "The Effect of Thermodynamic Constraints Upon the Mie-Grüneisen Equation of State." In *Constitutive Laws*, A.M. Rajendran and R.C. Batra (eds.), CIMNE: Barcelona, pp. 46–51, 1995.
- 19. Brillouin, L. Wave Propagation in Periodic Structures. New York: Dover, 1953. Originally published by New York: McGraw Hill, 1946.
- Kerley, G.I. "User Manual for PANDA II: A Computer Code for Calculating Equations of State." SAND88-2291, Sandia National Laboratories, Albuquerque, NM, July, 1991.
- 21. Mao, H.K., P.M. Bell, J.W. Shaner, and D.J. Steinberg. "Specific Volume Measurements of Cu, Mo, Pd, and Ag and Calibration of the Ruby R₁ Fluoresence Pressure Gauge From 0.06 to 1 Mbar." Journal of Applied Physics, 49 (6), pp. 3276–3283, 1978.
- 22. Molodets, A.M. "Grüneisen Function and the Zero-Temperature Isotherm of Three Metals up to Pressures of 10 TPa." *JETP*, **80** (3), pp. 467–471, 1995.

NO. OF COPIES ORGANIZATION

- 2 DEFENSE TECHNICAL INFO CTR ATTN DTIC DDA 8725 JOHN J KINGMAN RD STE 0944 FT BELVOIR VA 22060-6218
- 1 HQDA
 DAMO FDQ
 ATTN DENNIS SCHMIDT
 400 ARMY PENTAGON
 WASHINGTON DC 20310-0460
- 1 US MILITARY ACADEMY
 MATH SCI CTR OF EXCELLENCE
 DEPT OF MATHEMATICAL SCI
 ATTN MDN A MAJ DON ENGEN
 THAYER HALL
 WEST POINT NY 10996-1786
- 1 DIRECTOR
 US ARMY RESEARCH LAB
 ATTN AMSRL CS AL TP
 2800 POWDER MILL RD
 ADELPHI MD 20783-1145
- 1 DIRECTOR
 US ARMY RESEARCH LAB
 ATTN AMSRL CS AL TA
 2800 POWDER MILL RD
 ADELPHI MD 20783-1145
- 3 DIRECTOR
 US ARMY RESEARCH LAB
 ATTN AMSRL CI LL
 2800 POWDER MILL RD
 ADELPHI MD 20783-1145

ABERDEEN PROVING GROUND

2 DIR USARL ATTN AMSRL CI LP (305)

NO. OF COPIES	ORGANIZATION	NO. OF COPIES	ORGANIZATION
1	US ARMY DUSA OPS RSCH ATTN DANIEL WILLARD 102 ARMY PENTAGON WASHINGTON DC 20310-0102	1	COMMANDER USA STRATEGIC DEFNS CMD ATTN CSSD H LL T CROWLES HUNTSVILLE AL 35807-3801
1	ARMY SCIENCE BD WASH DC ATTN RICHARD L HALEY 2501 DUXBURY PLACE ALEXANDRIA VA 22308	2	COMMANDER US ARMY MICOM ATTN AMSMI RD ST WF D LOVELACE M SCHEXNAYDER
1	ARMY SCIENCE BOARD ORNL HEALTH SCI RSCH DIV ATTN ANETTA P WATSON BLDG 45005 MS 6101 OAK RIDGE TN 37831-6101	1	REDSTONE ARSENAL AL 35898-5250 MIS DEFNS & SPACE TECHNOLOGY ATTN CSSD SD T KENNETH H JORDAN PO BOX 1500 HUNTSVILLE AL 34807-3801
1	MIT LINCOLN LAB ARMY SCIENCE BOARD ATTN WADE M KORNEGAY 244 WOOD ST RM S2 139 LEXINGTON MA 02173	4	COMMANDER US ARMY BELVOIR RD&E CTR ATTN STRBE NAE B WESTLICH STRBE JMC T HANSHAW STRBE NAN
5	DEFENSE NUCLEAR AGENCY ATTN MAJ JAMES LYON CDR KENNETH W HUNTER TONY FREDERICKSON R JEFFREY LAWRENCE SPSP KIM KIBONG 6801 TELEGRAPH RD	3	S G BISHOP J WILLIAMS FORT BELVOIR VA 22060-5166 COMMANDER US ARMY RESEARCH OFFICE ATTN J CHANDRA
3	ALEXANDRIA VA 22310-3398 COMMANDER US ARMY ARDEC ATTN AMSTA AR FSA E W P DUNN J PEARSON	1	K IYER J BAILEY PO BOX 12211 RESEARCH TRIANGLE PARK NC 27709-2211 NAVAL RESEARCH LABORATORY ATTN ANDREW E WILLIAMS CODE 6684
1	E BAKER PICATINNY ARSENAL NJ 07806-5000 COMMANDER US ARMY ARDEC ATTN AMSTA AR CCH V M D NICOLICH PICATINNY ARSENAL NJ 07806-5000	1	4555 OVERLOOK AVE SW WASHINGTON DC 20375 NAVAL AIR WARFARE CTR ATTN STEPHEN A FINNEGAN BOX 1018 RIDGECREST CA 93556
1	COMMANDER US ARMY ARDEC ATTN E ANDRICOPOULOS PICATINNY ARSENAL NJ 07806-5000	3	COMMANDER NAVAL WEAPONS CENTER ATTN T T YEE CODE 3263 D THOMPSON CODE 3268 W J MCCARTER CODE 6214 CHINA LAKE CA 93555

- 9 COMMANDER
 NAVAL SURFACE WARFARE CTR
 DAHLGREN DIVISION
 ATTN C R GARRETT CODE G22
 W HOLT CODE G22
 CHARLES R ELLINGTON
 W WALLACE MORTON JR
 JOHN M NELSON
 WILLIAM J STROTHER
 D L DICKINSON CODE G24
 L F WILLIAMS CODE G33
 M J SILL CODE H11
 17320 DAHLGREN RD
 DAHLGREN VA 22448-5000
- 13 COMMANDER NAVAL SURFACE WARFARE CTR ATTN D TASKER CODE 9220 W WILSON P C HUANG CODE G402 B A BAUDLER CODE R12 R H MOFFETT CODE R12 R GARRETT CODE R12 T L JUNGLING CODE R32 R DAMINITY CODE U43 J P MATRA P WALTER L MENSI K KIDDY F J ZERILLI 10901 NEW HAMPSHIRE AVE SILVER SPRING MD 20903-5000
- 1 DIRECTOR
 NAVAL CIVIL ENGRNG LAB
 ATTN J YOUNG CODE L56
 PORT HUENEME CA 93043
- 1 NAVAL POST GRADUATE SCHOOL PHYSICS DEPARTMENT ATTN JOSEPH STERNBERG MONTEREY CA 93943

- 5 AIR FORCE ARMAMENT LAB
 ATTN AFATL DLJW
 W COOK
 M NIXON
 AFATL DLJR J FOSTER
 AFATL MNW
 LT D LOREY
 R D GUBA
 EGLIN AFB FL 32542
- 5 WRIGHT LABS
 ATTN MNMW JOEL W HOUSE
 ARMAMENT DIRECTORATE STE 326 B1
 RONALD D HUNT
 BRYAN MILLIGAN
 BRUCE C PATTERSON
 WADE H VAUGHT
 101 W EGLIN BLVD
 EGLIN AFB FL 32542-6810
- 2 USAF PHILLIPS LABORATORY
 ATTN PL WSCD FIROOZ ALLAHDADI
 PV VTA DAVID SPENCER
 3550 ABERDEEN AVE SE
 KIRTLAND AFB NM 87117-5776
- 1 USAF PHILLIPS LABORATORY VTSI ATTN ROBERT ROYBAL KIRTLAND AFB NM 87117-7345
- 1 AFIT ENC ATTN DAVID A FULK WRIGHT PATTERSON AFB OH 45433
- 1 US DEPT OF ENERGY
 IDAHO OPERATIONS OFFICE
 ATTN RONALD H JOHNSON
 . 850 ENERGY DR
 IDAHO FALLS ID 83401-1563

NO. OF COPIES	ORGANIZATION	NO. OF COPIES	ORGANIZATION
44	DIRECTOR SANDÍA NATL LABS ATTN E S HERTEL JR MS 0819 L N KMETYK E H BARSIS MS 031 ERIC W REECE MS 0307 DANIEL P KELLY MS 0307 L WEIRICK MS 0327 R TACHAU MS 0425 D LONGCOPE MS 0439 R REEDER D HAYES MS 0457 J ASAY MS 0458 W TEDESCHI MS 0482 J SCHULZE MS 0483 J SOUTHWARD PAUL A LONGMIRE MS 0560 T TRUCANO MS 0819 J MICHAEL MCGLAUN MS 0819 R BRANNON MS 0820 J ANG MS 0821 M BOSLOUGH MS 0821 L CHHABILDAS MS 0821 D CRAWFORD MS 0821 J DICK M FURNISH MS 0821 C HALL MS 0821 W REINHART MS 0821 P STANTON MS 0821 P TAYLOR ORG 1432 D KERNAN ORG 1433 C KONRAD K LANG M KIPP DIV 1533 A ROBINSON DIV 1533 P YARRINGTON DIV 1533 J MCGLAWA DIV 1541 M FORRESTAL DIV 1551 R GRAHAM DIV 1551 R GRAHAM DIV 1551 C HILLS DIV 1822 W J ANDRZEJEWSKI DIV 2512 D MARCHI DIV 2512 W VANDERMOLEN ORG 2653 B LEVIN ORG 7816 R O NELLUMS DIV 9122 PO BOX 5800 ALBUQUERQUE NM 87185-0307	3	DIRECTOR LLNL ATTN JOSEPH V REPA MS A133 MICHAEL O SCHNICK MS F607 EDWARD J CHAPYAK MS F664 PO BOX 1663 LOS ALAMOS NM 87545 DIRECTOR LLNL ATTN R E TIPTON L35 D BAUM L35 T MCABEE MS 35 M J MURPHY R PIERCE L122 R ROSINKY L122 O J ALFORD L122 D STEWART L122 T VIDLAK L122 B R BOWMAN L122 W DIXON L122 A C MITCHELL J A MORIARTY R A HEINLE N C HOLMES R PERRET L163 W SHOTTS L163 H KRUGER L178 G POMYKAL L178 M SHANNON G SIMONSON L180 A SPERO L180 W TAO L282 P URTIEW L282 A HOLT L290 J E REAUGH L290 W J NELLIS L299 D WOOD L352 D GAVEL L495 J HUNTER L495 E JOHANSSON L495 R M KUKLO L874 G W REPP PO BOX 808 LIVERMORE CA 94550

NO. OF		NO. OF	
COPIES	ORGANIZATION	COPIES	ORGANIZATION
8	DIRECTOR	12	DIRECTOR
	LLNL		LANL
	ATTN DOUGLAS R FAUX L125		ATTN P HOWE MS P915
	NORMAN W KLINO L125H		W DEAL MS P915
	ROBERT BARKER L159		J KENNEDY MS P915
	MILTON FINGER L163		A ROACH MS P915
	MICHAEL GERASSIMENKO L178		L SCHWALBE
	FRANK A HANDLER L182		W HEMSING MS P940
	STEPHEN G COCHRAN L389		E POGUE MS P940
	BMDO ROBERT M HALL		J MCAFEE MS P950
	PO BOX 808		D PAISLEY MS P950
	LIVERMORE CA 94551-0808		L PICKLESIMER MS P950
			R WARNES MS P950
33	DIRECTOR		S SHEFFIELD MS P952
	LANL		PO BOX 1663
	ATTN M LUCERO MS A105		LOS ALAMOS NM 87545
	D MANDELL	•	THE OF THE MATTER AND DOLL OF TOOL
	K MARK	3	ENERGETIC MATERIALS RSCH CTR/DOE NEW MEXICO INST OF MINING & TECH
	J P RITCHIE MS B214 T14		
	R DINGUS MS B218		ATTN DAVID J CHAVEZ LARRY LIBERSKY
	N KRIKORIAN MS B228 R KIRKPATRICK MS B229		FRED SANDSTROM
	R THURSTON MS B229		CAMPUS STATION
	C T KLINGNER MS B294		SOCORRO NM 87801
	R MILLER MS B294		50C0R(0 14M 07001
	S J MOSSO	1	NASA
	B SHAFER MS C931	•	LEWIS RESEARCH CENTER
	G GISLER MS D436		ATTN J FERRANTE
	C RAGAN MS D449		CLEVELAND OH 44135
	B LAUBSCHER MS D460		
	R WELLS MS F607	3	NASA
	R KOPP MS F645		JOHNSON SPACE CENTER
	R STELLINGWERF MS F645		ATTN ERIC CHRISTIANSEN
	C WINGATE MS F645		JEANNE LEE CREWS
	T ADAMS MS F663		FREDRICH HORZ
	R GODWIN MS F663		MAIL CODE SN3
	K JACOBY MS F663		2101 NASA RD 1
	W SPARKS MS F663		HOUSTON TX 77058
	J SHANER MS F670	1	ADDITION DESCRIPTION I AD
	G CANAVAN MS F675	1	APPLIED RESEARCH LAB
	R GREINER MS G740		ATTN JEFFREY A COOK
	J HILLS MS G770 B HOGAN MS G770		10000 BURNETT ROAD AUSTIN TX 78758
	J BOLSTAD MS G787		AUSTIN 1A 78738
	J WALSH MS G787	1	GM RESEARCH LABS
	R DAVIDSON MS K557	1	ATTN J R SMITH
	R HENNINGER MS K557 N6		WARREN MI 48090
	T ROLLET MS K574		77 HGG11 III 10070
	PO BOX 1663		·
	LOS ALAMOS NM 87545		
	200 122 2100 1414 010 10		

NO. OF COPIES	ORGANIZATION	NO. OF COPIES	ORGANIZATION
5	JET PROPULSION LABORATORY ATTN ZDENEK SEKANINA PAUL WEISSMAN BOB WEST JAMES ZWISSLER MARC ADAMS IMPACT PHYSICS GROUP 4800 OAK GROVE DR PASADENA CA 91109	5	JOHNS HOPKINS UNIV APPLIED PHYSICS LAB ATTN TERRY R BETZER ALVIN R EATON RICHARD H KEITH DALE K PACE ROGER L WEST JOHNS HOPKINS ROAD LAUREL MD 20723
1	MAXWELL LABS S CUBED DIVISION ATTN GERALD A GURTMAN PO BOX 1620 LA JOLLA CA 92037	1	LOUISIANA STATE UNIVERSITY ATTN ROBERT W COURTER 948 WYLIE DR BATON ROUGE LA 70808
1	AMES LABORATORY IOWA STATE UNIVERSITY ATTN J H ROSE AMES IA 50011	1	MIT DEPT OF EARTH ATMOS AND PLANETARY SCIENCES ATTN HEIDI B HAMMELL 54 316 CAMBRIDGE MA 02139
1	CALIFORNIA ST UNIV NORTHRIDGE ATTN JON H SHIVELY 10343 4TH ST NW	1	NC STATE UNIVERSITY ATTN YASUYUKI HORIE RALEIGH NC 27695-7908
2	ALBUQUERQUE NM 87114 CALTECH	1	PENNSYLVANIA STATE UNIVERSITY ATTN PHYSICS DEPT UNIVERSITY PARK PA 16802
	ATTN ANDREW P INGERSOLL MS 170 25 THOMAS J AHRENS MS 252 21 1201 E CALIFORNIA BLVD PASADENA CA 91125	5	SOUTHWEST RESEARCH INSTITUTE ATTN C ANDERSON D LITTLEFIELD S A MULLIN
1	CALTECH ATTN GLENN ORTON MS 169 237 4800 OAK GROVE DR PASADENA CA 91007		B COUR PALAIS J WALKER PO DRAWER 28510 SAN ANTONIO TX 78284
1	DREXEL UNIVERSITY ATTN PHYSICS DEPT 32ND & CHESTNUT ST PHILADELPHIA PA 19104	1	STANFORD UNIVERSITY ATTN BRIAN PETERSEN RAINS 16L 704 CAMPUS DR STANFORD CA 94305
1	GEORGIA INSTITUTE OF TECHNOLOGY COMPUTATIONAL MODELING CENTER ATTN S ATLURI ATLANTA GA 30332-0356	1	TEXAS A&M UNIVERSITY PHYSICS DEPARTMENT ATTN DAN BRUTON COLLEGE STATION TX 77843-4242

- 1 UC DAVIS
 INST OF THEORETICAL DYNAMICS
 ATTN E G PUCKETT
 DAVIS CA 95616
- 1 UC SANTA BARBARA INSTITUTE FOR THEORETICAL PHYSICS ATTN F GUINEA SANTA BARBARA CA 93106
- 2 UNIV OF ALA HUNTSVILLE AEROPHYSICS RSCH CTR ATTN GARY HOUGH DAVID J LIQUORNIK PO BOX 999 HUNTSVILLE AL 35899
- 1 UNIV OF ALA HUNTSVILLE CIVIL ENGRNG DEPT ATTN WILLIAM P SCHONBERG HUNTSVILLE AL 35899
- 1 UNIV OF ALABAMA ATTN A EUGENE CARDEN 86 EASTERN HILLS COTTONDALE AL 35453
- 1 UNIVERSITY OF ARIZONA ATTN DAVID H LEVY TUCSON AZ 85721
- 1 UNIVERSITY OF CA BERKLEY MECHNL ENGRNING DEPT GRADUATE OFFICE ATTN KEZHUN LI BERKELEY CA 94720
- 1 UNIVERSITY OF CHICAGO
 DEPT OF THE GEOPHYSICAL SCIENCES
 ATTN G H MILLER
 5734 S ELLIS AVE
 CHICAGO IL 60637
- 1 UNIVERSITY OF COLORADO ATTN T MACLAY CAMPUS BOX 431 NNT 341 BOULDER CO 80309

- 4 UNIVERSITY OF DELAWARE
 DEPT OF MECHANICAL ENGINEERING
 ATTN PROF J GILLESPIE
 DEAN R B PIPES
 PROF J VINSON
 PROF D WILKINS
 NEWARK DE 19716
- 1 UNIVERSITY OF MARYLAND ATTN PHYSICS DEPT (BLDG 082) COLLEGE PARK MD 20742
- 1 UNIVERSITY OF PUERTO RICO DEPT CHEMICAL ENGINEERING ATTN L A ESTEVEZ MAYAGUEZ PR 00681-5000
- 1 UNIVERSITY OF TEXAS
 DEPT OF MECHANICAL ENGINEERING
 ATTN ERIC P FAHRENTHOLD
 AUSTIN TX 78712
- 1 VIRGINIA POLYTECHNIC INSTITUTE COLLEGE OF ENGINEERING ATTN R BATRA BLACKSBURG VA 24061-0219
- 1 ADVANCED TECHNOLOGY INC ATTN J ADAMS PO BOX 125 DAHLGREN VA 22448-0125
- 2 AEROJET ELECTRO SYS CO ATTN WARHEAD SYSTEMS J CARLEONE S KEY PO BOX 296 AZUSA CA 91702
- 2 AEROJET ORDNANCE ATTN P WOLF G PADGETT 1100 BULLOCH BLVD SOCORRO NM 87801

- 3 ALLIANT TECHSYSTEMS INC
 ATTN T HOLMQUIST MN11 2720
 R STRYK
 G R JOHNSON MN11 2925
 600 SECOND ST NE
 HOPKINS MN 55343
- 1 ALME AND ASSOCIATES ATTN MARVIN L ALME 6219 BRIGHT PLUME COLUMBIA MD 21044-3790
- 1 APPLIED RESEARCH ASSOC INC ATTN JEROME D YATTEAU 5941 S MIDDLEFIELD RD SUITE 100 LITTLETON CO 80123
- 1 APPLIED RESEARCH ASSOC INC ATTN FRANK MAESTAS 4300 SAN MATEO BLVD SE ALBUQUERQUE NM 87110
- 1 BATTELLE
 ATTN ROBER M DUGAS
 7501 S MEMORIAL PKWY SUITE 101
 HUNTSVILLE AL 35802-2258
- 3 BATTELLE
 ATTN S GOLASKI
 R JAMESON
 TECHNICAL LIBRARY
 505 KING AVE
 COLUMBUS OH 43201
- 3 BOEING AEROSPACE CO
 SHOCK PHYSICS & APPLIED MATH
 ENGINEERING TECHNOLOGY
 ATTN R HELZER
 T MURRAY
 J SHRADER
 PO BOX 3999
 SEATTLE WA 98124
- 1 BOEING HOUSTON SPACE STN ATTN RUSSELL F GRAVES BOX 58747 HOUSTON TX 77258

- BRIGS CO
 ATTN JOSEPH E BACKOFEN
 2668 PETERSBOROUGH ST
 HERNDON VA 22071-2443
- 3 CALIFORNIA RSCH & TECHNOLOGY
 ATTN R BROWN
 R FRANZEN
 D ORPHAL
 5117 JOHNSON DR
 PLEASANTON CA 94566
- CALIFORNIA RSCH & TECHNOLOGY
 ATTN M MAJERUS
 PO BOX 2229
 PRINCETON NJ 08543
- COMPUTATIONAL MECHANICS
 CONSULTANTS
 ATTN J A ZUKAS
 PO BOX 11314
 BALTIMORE MD 21239-0314
- DEFENSE TECHNOLOGY INTERNATIONAL INC ATTN D E AYER THE STARK HOUSE 22 CONCORD ST NASHUA NH 03060
- 1 DESKIN RESEARCH GROUP INC ATTN EDWARD COLLINS 2270 AGNEW RD SANTA CLARA CA 95054
- 2 EI DUPONT DE NEMOURS & CO
 ATTN L MINOR
 B SCOTT
 SECURITY DIR LEGAL DEPARTMENT
 PO BOX 1635
 WILMINGTON DE 19899
- 1 G E DUVALL 5814 NE 82ND COURT VANCOUVER WA 98662-5944

NO. OF COPIES ORGANIZATION 3 DYNA FAST COI

- 3 DYNA EAST CORP ATTN P C CHOU R CICCARELLI W FLIS 3620 HORIZON DRIVE KING OF PRUSSIA PA 19406
- 3 DYNASEN
 ATTN JACQUES CHAREST
 MICHAEL CHAREST
 MARTIN LILLY
 20 ARNOLD PL
 GOLETA CA 93117
- 1 ELORET INSTITUTE
 ATTN DAVID W BOGDANOFF MS 230 2
 NASA AMES RESEARCH CENTER
 MOFFETT FIELD CA 94035
- 3 ENIG ASSOCIATES INC
 ATTN J ENIG
 D J PASTINE
 M COWPERTHWAITE
 SUITE 500
 11120 NEW HAMPSHIRE AVE
 SILVER SPRING MD 20904-2633
- 1 EXPLOSIVE TECHNOLOGY
 ATTN M L KNAEBEL
 PO BOX KK
 FAIRFIELD CA 94533
- 1 GB TECH LOCKHEED
 ATTN JAY LAUGHMAN
 2200 SPACE PARK SUITE 400
 HOUSTON TX 77258
- 2 GB TECH LOCKHEED
 ATTN LUCILLE BORREGO C23C
 JOE FALCON JR C23C
 2400 NASA ROAD 1
 HOUSTON TX 77058
- 2 GENERAL RESEARCH CORP ATTN A CHARTERS T MENNA PO BOX 6770 SANTA BARBARA CA 93160-6770

- 2 GRC INTERNATIONAL
 ATTN TIMOTHY M CUNNINGHAM
 WILLIAM M ISBELL
 5383 HOLLISTER AVE
 SANTA BARBARA CA 93111
- INST OF ADVANCED TECHNOLOGY
 UNIVERSITY OF TX AUSTIN
 ATTN STEPHEN J BLESS
 JAMES CAZAMIAS
 HARRY D FAIR
 THOMAS M KIEHNE
 NICK LYNCH
 4030-2 W BRAKER LN
 AUSTIN TX 78759
- 1 KAMAN SCIENCES CORP ATTN DENNIS L JONES 2560 HUNTINGTON AVE SUITE 200 ALEXANDRIA VA 22303
- 8 KAMAN SCIENCES CORP
 ATTN J ELDER
 RICHARD P HENDERSON
 DAVID A PYLES
 FRANK R SAVAGE
 JAMES A SUMMERS
 JAMES S WILBECK
 TIMOTHY W MOORE
 THY YEM
 600 BLVD S SUITE 208
 HUNTSVILLE AL 35802
- 3 KAMAN SCIENCES CORP
 ATTN SHELDON JONES
 GARY L PADEREWSKI
 ROBERT G PONZINI
 1500 GRDN OF THE GODS RD
 COLORADO SPRINGS CO 80907
- 4 KAMAN SCIENCES CORP
 ATTN NASIT ARI
 STEVE R DIEHL
 WILLIAM DOANE
 VERNON M SMITH
 PO BOX 7463
 COLORADO SPRINGS CO 80933-7463

- D R KENNEDY & ASSOC INC ATTN D KENNEDY PO BOX 4003 MOUNTAIN VIEW CA 94040
- 1 KERLEY PUBLISHING SERVICES ATTN G I KERLEY PO BOX 13835 ALBUQUERQUE NM 87192-3835
- 2 KTECH CORPORATION
 ATTN FRANK W DAVIES
 LARRY M LEE
 901 PENNSYLVANIA NE
 ALBUOUEROUE NM 87110
- 1 LIVERMORE SOFTWARE TECH CORP ATTN J O HALLQUIST 2876 WAVERLY WAY LIVERMORE CA 94550
- 1 LOCKHEED ENGRNG & SPC SCIENCES ATTN E CYKOWSKI MS B22 2400 NASA RD 1 HOUSTON TX 77058
- 1 LOCKHEED MARTIN MISSLE & SPACE ATTN WILLIAM R EBERLE PO BOX 070017 HUNTSVILLE AL 35807
- 3 LOCKHEED MARTIN MISSILE & SPACE
 ATTN M A LEVIN ORG 81 06 BLDG 598
 M R MCHENRY
 T A NGO ORG 81 10 BLDG 157
 111 LOCKHEED WAY
 SUNNYVALE CA 94088
- 4 LOCKHEED MISSILE & SPACE CO ATTN JOHN R ANDERSON WILLIAM C KNUDSON S KUSUMI 0 81 11 BLDG 157 J PHILLIPS 0 54 50 PO BOX 3504 SUNNYVALE CA 94088

- 1 LOCKHEED MISSILE & SPACE CO ATTN R HOFFMAN SANTA CRUZ FACILITY EMPIRE GRADE RD SANTA CRUZ CA 95060
- 1 LOCKHEED NASA JSC SPACE SCIENCE BRANCH ATTN JAMES HYDE BOX 58561 MC B22 HOUSTON TX 77258
- 1 LOCKHEED MARTIN AEROSPACE ATTN D R BRAGG PO BOX 5837 MP 109 ORLANDO FL 32855
- 1 MCDONNELL DOUGLAS
 ASTRONAUTICS CO
 ATTN B L COOPER
 5301 BOLSA AVE
 HUNTINGTON BEACH CA 92647
- ORLANDO TECHNOLOGY INC ATTN DANIEL A MATUSKA PO BOX 855 SHALIMAR FL 32579
- 1 PHYSICAL SCIENCES INC ATTN PETER NEBOLSINE 20 NEW ENGLAND BUS CTR ANDOWER MA 01810
- 3 PHYSICS INTERNATIONAL
 ATTN R FUNSTON
 G FRAZIER
 L GARNETT
 PO BOX 5010
 SAN LEANDRO CA 94577
- 1 RAYTHEON ELECTRONIC SYSTEMS
 ATTN R KARPP
 50 APPLE HILL DRIVE
 TEWKSBURY MA 01876
- 1 ROCKWELL INTERNATIONAL ROCKETDYNE DIVISION ATTN J MOLDENHAUER 6633 CANOGA AVE HB 23 CANOGA PARK CA 91303

- 1 ROCKWELL INTERNATIONAL ROCKETDYNE DIVISION ATTN H LEIFER 16557 PARK LN CIRCLE LOS ANGELES CA 90049
- 1 ROCKWELL MISSILE SYS DIV ATTN T NEUHART 1800 SATELLITE BLVD DULUTH GA 30136
- 1 S CUBED
 ATTN R SEDGWICK
 PO BOX 1620
 LA JOLLA CA 92038-1620
- 2 SAIC
 ATTN JAMES FURLONG
 GREGORY J STRAUCH
 1710 GOODRIDGE DR
 MCLEAN VA 22102
- 1 SAIC
 ATTN MICHAEL W MCKAY
 10260 CAMPUS POINT DR
 SAN DIEGO CA 92121
- 1 SHOCK TRANSIENTS INC ATTN DAVID DAVISON BOX 5357 HOPKINS MN 55343
- 2 SIMULATION & ENG CO INC ATTN ELSA I MULLINS STEVEN E MULLINS 8840 HWY 20 SUITE 200 N MADISON AL 35758
- 2 SOUTHERN RESEARCH INSTITUTE ATTN LINDSEY A DECKARD DONALD P SEGERS PO BOX 55305 BIRMINGHAM AL 35255-5305
- 3 SRI INTERNATIONAL
 ATTN JAMES D COLTON
 DR L SEAMAN
 DONALD A SHOCKLEY
 333 RAVENSWOOD AVE

NO. OF COPIES ORGANIZATION

MENLO PARK CA 94025

- 2 TELEDYNE BROWN ENGR ATTN JIM W BOOTH MARTIN B RICHARDSON PO BOX 070007 MS 50 HUNTSVILLE AL 35807-7007
- 1 ZERNOW TECHNICAL SVCS INC ATTN LOUIS ZERNOW 425 W BONITA AVE SUITE 208 SAN DIMAS CA 91773

ABERDEEN PROVING GROUND

26

DIR, USARL ATTN: AMSRL-WM, I MAY AMSRL-WM-T, W F MORRISON AMSRL-WM-TA. W GILLICH W BRUCHEY J DEHN **G FILBEY** W A GOOCH H W MEYER E J RAPACKI AMSRL-WM-TB, R FREY J STARKENBERG AMSRL-WM-TC. W S DE ROSSET T W BJERKE R COATES F GRACE K KIMSEY M LAMPSON D SCHEFFLER S SCHRAML **B SORENSEN** R SUMMERS W WALTERS AMSRL-WM-PB, A ZIELINSKI AMSRL-WM-PD, G GAZONAS

AMSRL-WM-WD, A PRAKASH

AMSRL-WM-MF, A RAJENDRAN

NO. OF

COPIES ORGANIZATION

13 DIR, USARL

ATTN: AMSRL-WM-TD,

A M DIETRICH

K FRANK

J HARRISON

M RAFTENBERG

G RANDERS-PEHRSON

J SANTIAGO

M SCHEIDLER

S SCHOENFELD

S SEGLETES (3 CP)

J WALTER

T WRIGHT

1 ABTEILUNG FUER PHYSIKALISCHE CHEMIE MONTANUNIVERSITAET ATTN E KOENIGSBERGER

A 8700 LEOBEN AUSTRIA

- 2 AWE
 ATTN MICHAEL GERMAN
 WAYNE HARRISON
 FOULNESS ESSEX SS3 9XE
 UNITED KINGDOM
- 1 BATTELLE INGENIEUTECHNIK GMBH DUESSELDORFFER STR 9 ESCHBORN D 65760 GERMANY
- 1 BHABHA ATOMIC RESEARCH CENTRE HIGH PRESSURE PHYSICS DIVISION ATTN N SURESH TROMBAY BOMBAY 400 085 INDIA
- BULGARIAN ACADEMY OF SCIENCES SPACE RESEARCH INSTITUTE ATTN VALENTIN GOSPODINOV 1000 SOFIA PO BOX 799 BULGARIA
- 1 CANADIAN ARSENALS LTD ATTN P PELLETIER 5 MONTEE DES ARSENAUX VILLIE DE GRADEUR PQ J5Z2 CANADA
- 1 CEA
 ATTN ROGER CHERET
 CEDEX 15
 313 33 RUE DE LA FEDERATION
 PARIS 75752
 FRANCE
- 1 CEA
 CISI BRANCH
 ATTN PATRICK DAVID
 CENTRE DE SACLAY BP 28
 GIF SUR YVETTE 91192
 FRANCE

NO. OF

COPIES ORGANIZATION

- 1 CEA/CESTA
 ATTN ALAIN GEILLE
 BOX 2 LE BARP 33114
 FRANCE
- 3 CENTRE D'ETUDES DE GRAMAT ATTN SOLVE GERARD CHRISTIAN LOUPIAS PASCALE OUTREBON GRAMAT 46500 FRANCE
- 2 CENTRE D'ETUDES DE LIMEIL-VALENTON ATTN CHRISTIAN AUSSOURD JEAN-CLAUDE BOZIER SAINT GEORGES CEDEX VILLENEUVE 94195 FRANCE
- 3 CENTRE D'ETUDES DE VAUJOURS ATTN PLOTARD JEAN-PAUL ERIC BOTTET TAT SIHN VONG BOITE POSTALE NO 7 COUNTRY 77181 FRANCE
- 1 CENTURY DYNAMICS LTD
 ATTN NIGEL FRANCIS
 12 CITY BUSINESS CTR
 BRIGHTON RD HORSHAM
 WEST SUSSEX RH1354BA
 UNITED KINGDOM
- 1 CRC RESEARCH INST INC
 ATTN MASAHIDE KATAYAMA
 STRUCTURAL ENGINEERING DEPT
 1 3 316 NAKASE MIHAMA KU
 CHIBA SHI 261 01
 JAPAN
- 1 DAT ETBS CETAM
 ATTN CLAUDE ALTMAYER
 ROUTE DE GUERRY BOURGES
 18015
 FRANCE

- 1 DEFENCE RESEARCH AGENCY ATTN W A J CARSON CHOBHAM LANE CHERTSEY SURREY KT16 0EE UNITED KINGDOM
- 1 DEFENCE RSCH ESTAB SUFFIELD ATTN D MACKAY RALSTON ALBERTA TOJ 2NO RALSTON CANADA
- DEFENCE RSCH ESTAB SUFFIELD
 ATTN CHRIS WEICKERT
 BOX 4000 MEDICINE HAT
 ALBERTA TIA 8K6
 CANADA
- 1 DEFENCE RSCH ESTAB VALCARTIER ATTN N GASS PO BOX 8800 COURCELETTE PQ GOA IRO CANADA
- 4 DEPARTMENTO DE QUIMICA FISICA
 FACULTAD DE CIENCIAS QUIMICAS
 UNIVERSIDAD COMPLUTENSE DE
 MADRID
 ATTN V G BAONZA
 M TARAVILLO
 M CACERAS
 J NUNEZ
 28040 MADRID
 SPAIN
- 1 DYNAMIC RESEARCH AB ATTN AKE PERSSON PARADISGRAND 7 SODERTALJE S 151 36 SWEDEN
- 4 ERNST MACH INSTITUT
 ATTN VOLKER HOHLER
 E SCHMOLINSKE
 E SCHNEIDER
 K THOMA
 ECKERSTRASSE 4
 D-7800 FREIBURG I BR 791 4
 GERMANY

- 1 ESTEC CS
 ATTN DOUGLAS CASWELL
 BOX 200 NOORDWIJK
 2200 AG
 NETHERLANDS
- 1 EUROPEAN SPACE AGENCY
 ATTN WALTER FLURY
 ESOC 5
 ROBT BOSCHT STRASSE
 DARMSTADT 6100
 GERMANY
- 2 EUROPEAN SPACE AGENCY ESTEC
 ATTN LUCY BERTHOUD
 MICHEL LAMBERT
 POSTBUS BOX 299 NOORDWIJK
 NL2200 AG
 NETHERLANDS
- FRENCH GERMAN RESEARCH INST
 ATTN HARTMUTH F LEHR
 ROLF HUNKLER
 ERICH WOLLMANN
 POSTFACH 1260
 WEIL AM RHEIN D-79574
 GERMANY
- 3 FRENCH GERMAN RESEARCH INST
 ATTN HANS-JURGEN ERNST
 FRANCIS JAMET
 PASCALE LEHMANN
 CEDEX 5 5 RUE DU GENERAL CASSAGNOU
 SAINT LOUIS 68301
 FRANCE
- HIGH ENERGY DENSITY RESEARCH CTR
 ATTN VLADIMIR E FORTOV
 GENADII I KANEL
 V A SKVORTSOV
 O YU VOJOBIEV
 IZHORSKAJA STR 13/19
 MOSCOW 127412
 RUSSIAN REPUBLIC

NO. OF

COPIES ORGANIZATION

- 5 IABG
 ATTN H J RAATSCHEN
 W SCHITTKE
 F F SCHARPPF
 M BORRMANN
 H G DORSCH
 EINSTEINSTRASSE 20
 D 8012 OTTOBRUN B MUENCHEN
 GERMANY
- 1 INSTITUTE FOR PROBLEMS OF STRENGTH ATTN G STEPANOV TIMIRYAZEVSKAYU STR 2 252014 KIEV UKRAINE
- 1 INSTITUTE OF CHEMICAL PHYSICS ATTN A YU DOLGOBORODOV KOSYGIN ST 4 V 334 MOSCOW
- 3 INSTITUTE OF CHEMICAL PHYSICS
 RUSSIAN ACADEMY OF SCIENCES
 ATTN A M MOLODETS
 S V RAZORENOV
 A V UTKIN
 142432 CHERNOGOLOVKA
 MOSCOW REGION
 RUSSIAN REPUBLIC
- 2 IOFFE PHYSICO TECHNICAL INSTITUTE
 DENSE PLASMA DYNAMICS
 LABORATORY
 ATTN EDWARD M DROBYSHEVSKI
 A KOZHUSHKO
 ST PETERSBURG 194021
 RUSSIAN REPUBLIC
- 1 IPE RAS
 ATTN A A BOGOMAZ
 DVORTSOVAIA NAB 18
 ST PETERSBURG
 RUSSIAN REPUBLIC
- 1 ISL
 ATTN CHANTERET PIERRE-YVES
 F68301 SAINT-LOUIS
 CEDEX 12 RUE DE I'INDUSTRIE
 BP 301
 FRANCE

NO. OF

COPIES ORGANIZATION

- 1 LABORATOIRE DE TECHNOLOGIE DES SURFACES ECOLE CENTRALE DE LYON ATTN PASCAL VINET BP 163 69131 ECULLY CEDEX FRANCE
- 2 LAVRENTYEV INST. HYDRODYNAMICS ATTN LEV A MERZHIEVSKY VICTOR V SILVESTROV NOVOSIBIRSK 630090 RUSSIAN REPUBLIC
- 1 MOSCOW INST OF PHYSICS & TECH ATTN S V UTYUZHNIKOV DEPT OF COMPUTATIONAL MATHEMATICS DOLGOPRUDNY 1471700 RUSSIAN REPUBLIC
- 1 NATL DEFENCE DENSITY RSCH CTR ATTN JOHAN FORSS FOA FOX 551 TUMBA S 14725 SWEDEN
- 3 NATL DEFENCE RESEARCH EST ATTN LARS HOLMBERG ULF LINDEBERG LARS GUNNAR OLSSON FOA BOX 551 TUMBA S 14725 SWEDEN
- 1 PML TNO
 ATTN EDWARD VAN RIET
 BOX 45
 RIJSWIJK 2230AA
 NETHERLANDS
- 1 PRB S A
 ATTN M VANSNICK
 AVENUE DE TERVUEREN 168 BTE 7
 BRUSSELS B 1150
 BELGIUM

NO. OF NO. OF COPIES ORGANIZATION COPIES ORGANIZATION TU CHEMNITZ-ZWICKAU 1 PRINS MAURITS LABORATORY 1 ATTN LOTHAR MEYER ATTN H J REITSMA FAKULTAET FUER MASCHINENBAU U. TNO BOX 45 VERFAHRENSTECHNIK RIJSWIJK 2280AA SCHEFFELSTRASSE 110 **NETHERLANDS** 09120 CHEMNITZ **GERMANY** 2 RAFAEL BALLISTICS CENTER ATTN EREZ DEKEL UK MINISTRY OF DEFENCE YEHUDA PARTOM ATTN GRAHAM J CAMBRAY PO BOX 2250 CBDE PORTON DOWN SALISBURY HAIFA 31021 WITTSHIRE SPR 0JQ ISRAEL UNITED KINGDOM RESEARCH INSTITUTE OF MECHANICS 1 UNIVERSIDAD DE CANTABRIA 1 NIZHNIY NOVGOROD STATE UNIVERSITY FACULTAD DE CIENCIAS ATTN A SADYRIN DEPARTMENTO DE FISICA APLICADA P.R. GAYARINA 23 KORP 6 ATTN J AMOROS NIZHNIY NOVGOROD 603600 AVDA DE LOS CASTROS S/N RUSSIAN REPUBLIC SANTANDER ROYAL ARMAMENT R&D ESTAB SPAIN 1 ATTN I CULLIS UNIVERSIDAD DE OVIEDO FORT HALSTEAD FACULTAD DE QUIMICA SEVENOAKS KENT TN14 7BJ DEPARTMENTO DE QUIMICA FISICA Y UNITED KINGDOM ANALITICA SAMARA STATE AEROSPACE UNIV ATTN E FRANCISCO 1 AVENIDA JULIAN CLAVERIA S/N ATTN L G LUKASHEV 33006 - OVIEDO SAMARA **SPAIN** RUSSIAN REPUBLIC UNIVERSITY OF GUELPH TECHNISCHE UNIVERSITAT MUENCHEN 1 PHYSICS DEPT ATTN EDUARD B IGENBERGS ATTN C G GRAY RICHARD WAGNER STR 18 111 **GUELPH ONTARIO** MUENCHEN 2 D8000 N1G 2W1 **GERMANY** CANADA TOMSK BRANCH OF THE INSTITUTE 1 UNIVERSITY OF KENT FOR STRUCTURAL MACROKINETICS UNIT FOR SPACE SCIENCES ATTN V GORELSKI ATTN PHILIPPE GENTA 8 LENIN SQ GSP 18 PAUL RATCLIFF TOMSK 634050 CANTERBURY KENT CT2 7NR RUSSIAN REPUBLIC

UNITED KINGDOM

Form Approved REPORT DOCUMENTATION PAGE OMB No. 0704-0188 Public reporting burden for this collection of information is estimated to average 1 hour per response, including the time for reviewing instructions, searching existing data sources, gethering and maintaining the data needed, and completing and reviewing the collection of information. Send comments regarding this burden estimate or any other aspect of this collection of information, including suggestions for reducing this burden, to Washington Headquarters Services, Directorate for Information Operations and Reports, 1215 Jefferson Davis Highway, Suits 1204, Arlington, VA 22202-4302, and to the Office of Management and Budget, Paperwork Reduction Project(0704-0188), Washington, DC 20503. 1. AGENCY USE ONLY // agree blank! 2. REPORT DATE 1. AGENCY USE ONLY (Leave blank) March 1997 Final, Sep - Oct 96 4. TITLE AND SUBTITLE 5. FUNDING NUMBERS On Theories of the Grüneisen Parameter PR: 1L162618AH80 6. AUTHOR(S) Steven B. Segletes and William P. Walters 8. PERFORMING ORGANIZATION 7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) REPORT NUMBER U.S. Army Research Laboratory ATTN: AMSRL-WM-TD ARL-TR-1303 Aberdeen Proving Ground, MD 21005-5066 10.SPONSORING/MONITORING 9. SPONSORING/MONITORING AGENCY NAMES(S) AND ADDRESS(ES) AGENCY REPORT NUMBER 11. SUPPLEMENTARY NOTES 12b. DISTRIBUTION CODE 12a, DISTRIBUTION/AVAILABILITY STATEMENT Approved for public release; distribution is unlimited. 13. ABSTRACT (Maximum 200 words) A recently proposed equation of state is further examined for compatibility with prior theory. It is found, by adopting a more realistic functional relationship for one of the thermodynamic state variables and eliminating higher-order terms from the model, that three historical theories for determination of the Grüneisen function fall directly out of the recently proposed model. One is led to conclude that the newer model captures the essence of all of the older theories, each of which being applicable under an appropriate circumstance. 15. NUMBER OF PAGES 14. SUBJECT TERMS equation of state, lattice frequency, Grüneisen, Debye temperature 16. PRICE CODE 20. LIMITATION OF ABSTRACT 18. SECURITY CLASSIFICATION 19. SECURITY CLASSIFICATION 17. SECURITY CLASSIFICATION OF THIS PAGE OF ABSTRACT OF REPORT **UNCLASSIFIED** UNCLASSIFIED UNCLASSIFIED

Standard Form 298 (Rev. 2-89) Prescribed by ANSI Std. 239-18 298INTENTIONALLY LEFT BLANK.

USER EVALUATION SHEET/CHANGE OF ADDRESS

This Laboratory undertakes a continuing effort to improve the quality of the reports it publishes. Your comments/answers to the items/questions below will aid us in our efforts. 1. ARL Report Number/Author <u>ARL-TR-1303 (Segletes)</u> Date of Report <u>March 1997</u> 2. Date Report Received ______ 3. Does this report satisfy a need? (Comment on purpose, related project, or other area of interest for which the report will 4. Specifically, how is the report being used? (Information source, design data, procedure, source of ideas, etc.) 5. Has the information in this report led to any quantitative savings as far as man-hours or dollars saved, operating costs avoided, or efficiencies achieved, etc? If so, please elaborate. 6. General Comments. What do you think should be changed to improve future reports? (Indicate changes to organization, technical content, format, etc.) Organization Name E-mail Name **CURRENT** ADDRESS Street or P.O. Box No. City, State, Zip Code 7. If indicating a Change of Address or Address Correction, please provide the Current or Correct address above and the Old or Incorrect address below.

Organization

Name

Street or P.O. Box No.

City, State, Zip Code

OLD

ADDRESS

(Remove this sheet, fold as indicated, tape closed, and mail.)
(DO NOT STAPLE)

DEPARTMENT OF THE ARMY

OFFICIAL BUSINESS



POSTAGE WILL BE PAID BY ADDRESSEE

DIRECTOR
US ARMY RESEARCH LABORATORY
ATTN AMSRL WM TD
ABERDEEN PROVING GROUND MD 21005-5066

NO POSTAGE NECESSARY IF MAILED IN THE UNITED STATES